

## Short-range order and microhardness of the compositionally complex alloy $\text{Al}_8\text{Co}_{17}\text{Cr}_{17}\text{Cu}_8\text{Fe}_{17}\text{Ni}_{33}$

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Introduction

Experimental

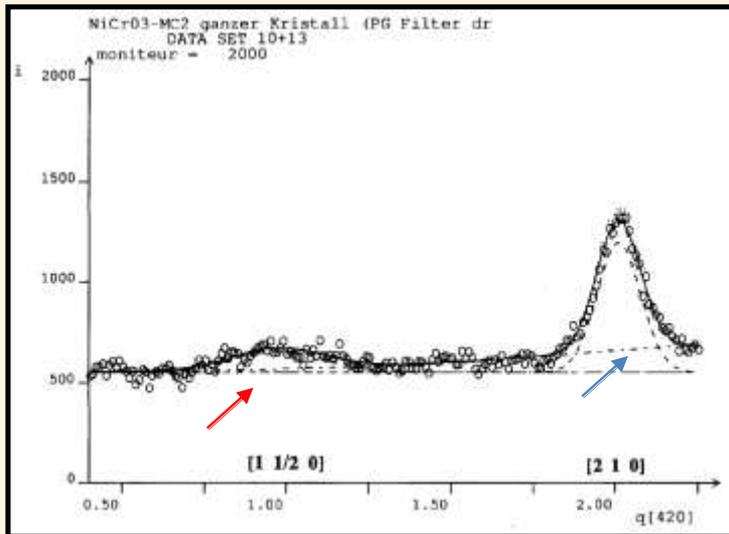
Results

- Soft X-ray absorption spectroscopy
- Extended X-ray absorption spectroscopy
- Hardness
- Molecular dynamic simulations

Conclusions and outlook

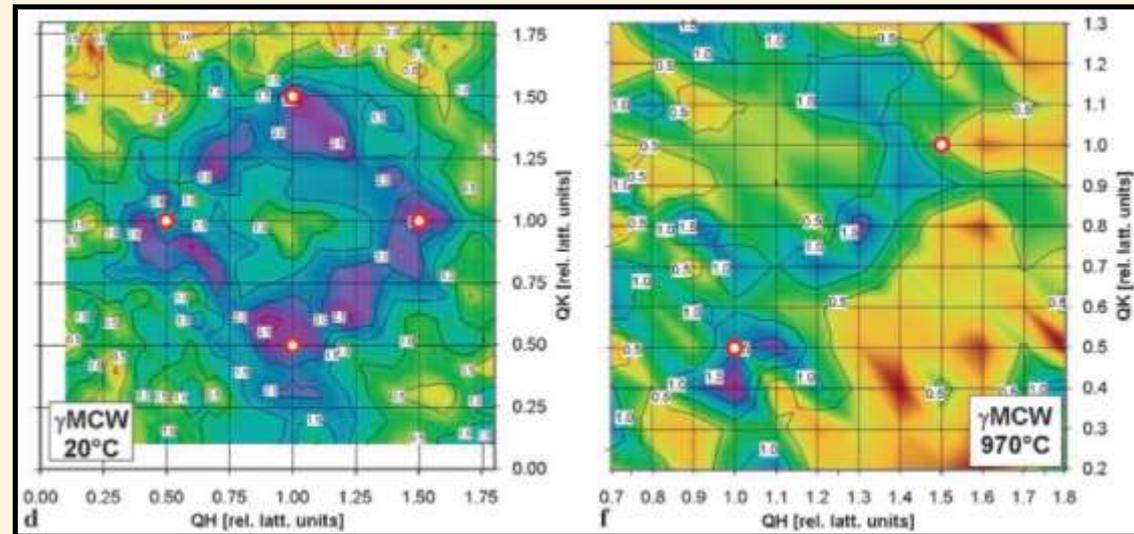
# Short-range order in superalloys

by neutron scattering



**MC-2**, RT

(Glas et al., Acta Mater 1996)



**MCW** at 20°C, 970°C

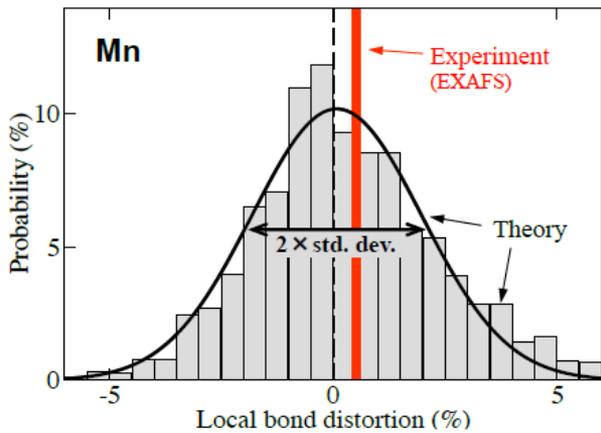
(Prem et al., Appl. Phys. 2002)

- Diffuse scattering hints of Short-Range Order
  - superstructure peak at  $[1 \frac{1}{2} 0]$
  - diffuse background at  $[2 1 0]$
- ~ 1.5 nm ordered domains at room temperature (MC-2)
- Short-Range Order dissolves above  $T = 600 \text{ } ^\circ\text{C}$  (MCW)

**Short-Range Order in Compositionally Complex Alloys?**

# Short-range order in compositionally complex alloys

(b) Locally resolved lattice distortions

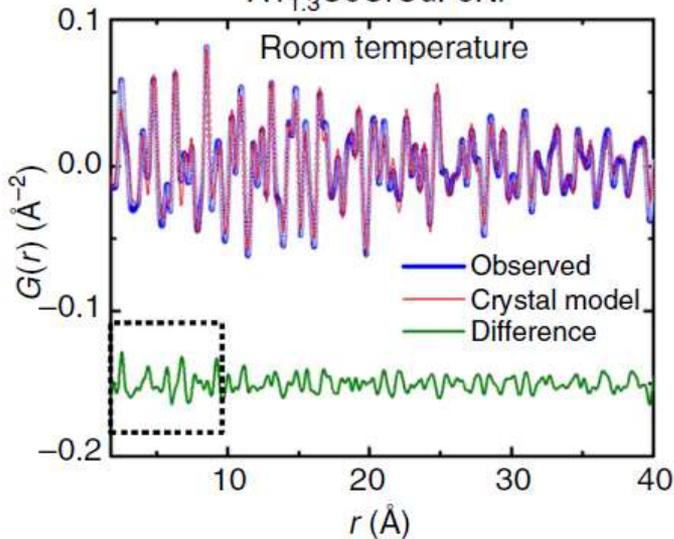


## FeCoNiCrMn

*Entropy 2016, 18, 321*

- Small element-resolved mean lattice distortions ( $\sim 0.1\%$ )
- High local fluctuations of lattice distortions ( $\sim 2-3\%$ )

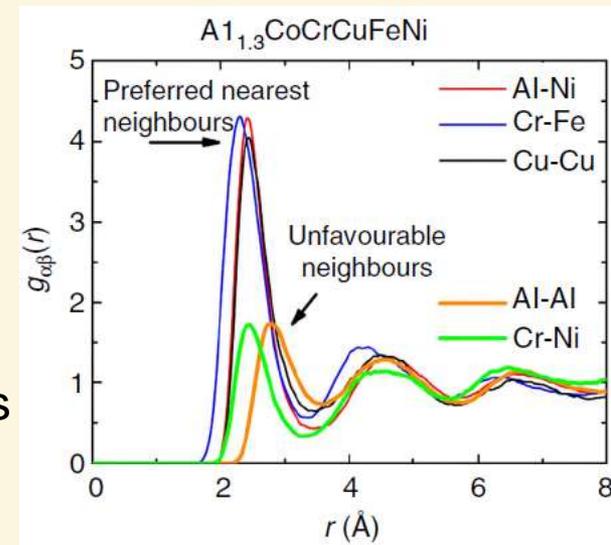
## Al<sub>1.3</sub>CoCrCuFeNi



## Al<sub>1.3</sub>CoCrCuFeNi

*Nat. Comm. 2015, 6, 5964*

- Structure  $< 10 \text{\AA}$  not well modelled by cubic fcc
- AIMD simulations show preferred nearest neighbors



X-ray absorption spectroscopy  
 $5 \text{ KeV} < E < 10 \text{ KeV}$

BESSYII: beamlines KMC-2 and KMC-3

- Mode: fluorescence
- Temperatures: 20 K – 300 K

Soft X-ray Absorption Spectroscopy  
 $100 \text{ eV} < E < 1500 \text{ eV}$

BESSYII: Russian-German beamline  
Mode: total electron yield (<10nm  
information depth) in ultra-high vacuum

Hardness measurements

- Vickers indentation
- Nanoindentation

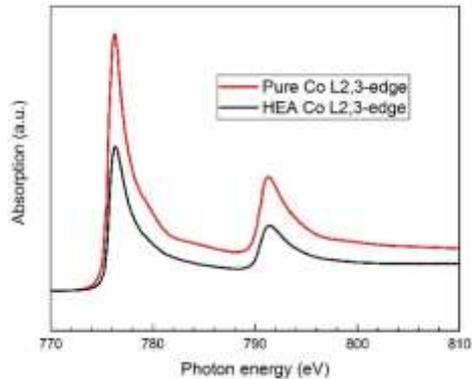
Ab-initio molecular dynamics  
simulations

Dr. Yao Liu (Shandong University)  
Vienna ab initio simulation package

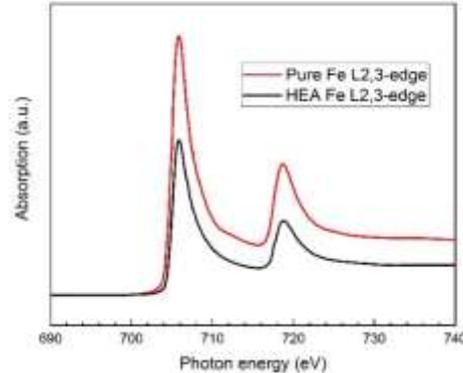
$\text{Al}_8\text{Co}_{17}\text{Cr}_{17}\text{Cu}_8\text{Fe}_{17}\text{Ni}_{33}$  specimens  
2-to-4 components precursors

Bulk (Single crystal, polycrystal)  
Splat

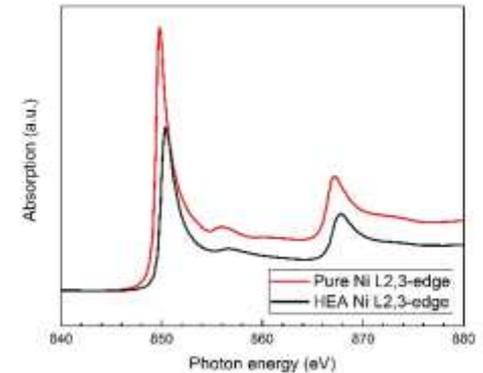
# Results: Soft X-ray absorption spectroscopy



L<sub>2,3</sub> absorption spectra of Co in pure Co (red line) and in CCA (black line)



L<sub>2,3</sub> absorption spectra of Fe in pure Fe (red line) and in CCA (black line)



L<sub>2,3</sub> absorption spectra of Ni in pure Ni (red line) and in CCA (black line)

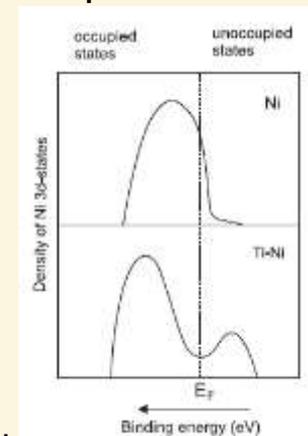
Spin-orbit splitting visible in all spectra ( $\Delta E(\text{Co}): \approx 15$  eV,  $\Delta E(\text{Fe}): \approx 13$  eV,  $\Delta E(\text{Ni}): \approx 17$  eV)

- Absorption spectrum of Co/Fe in CCA similar to absorption spectra of Co/Fe pure metals

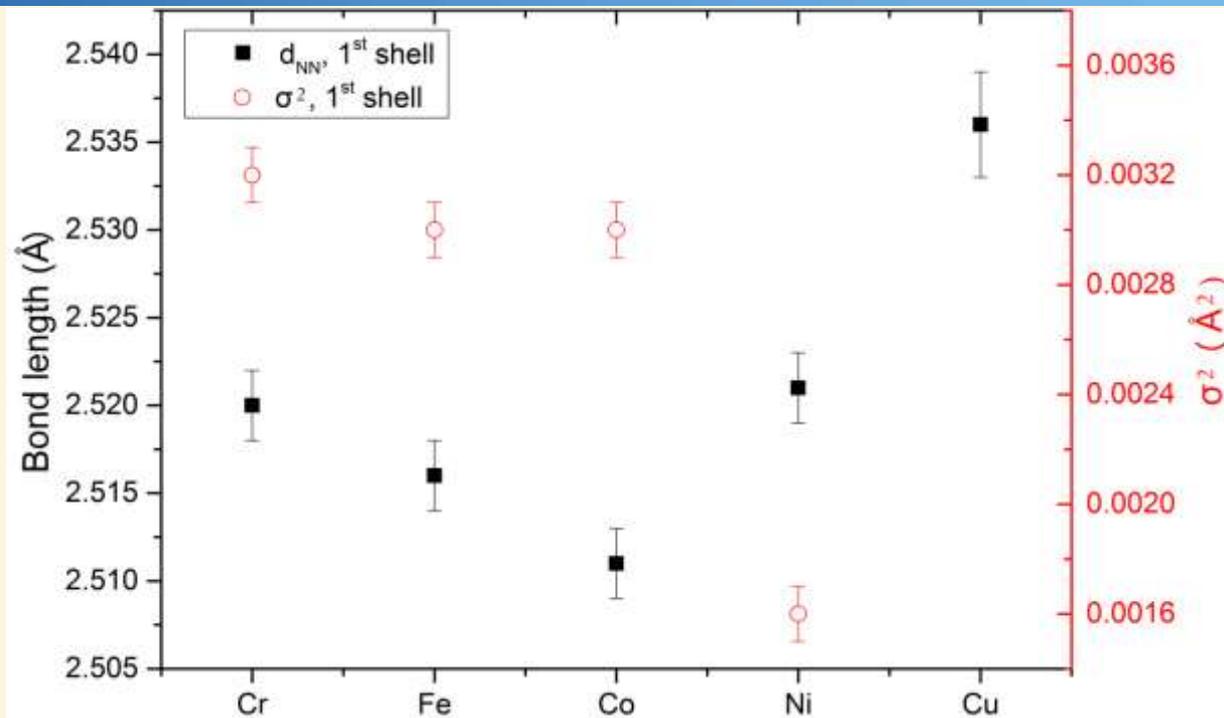
⇒ Electronic structure of Co/Fe in CCA similar to that of Co/Fe metals

- Absorption spectrum of Ni in CCA shifted to higher energy compared with Ni in Ni metal

⇒ Electronic structure of Ni in Ni **dissimilar** to Ni in CCA  
(i) higher binding energy of Ni 2p levels in CCA  
(ii) shift in density of empty Ni d-states



# Results: Extended X-ray Absorption Fine Structure



Al missing. Al-K edge logistically difficult to measure (few beamlines available)

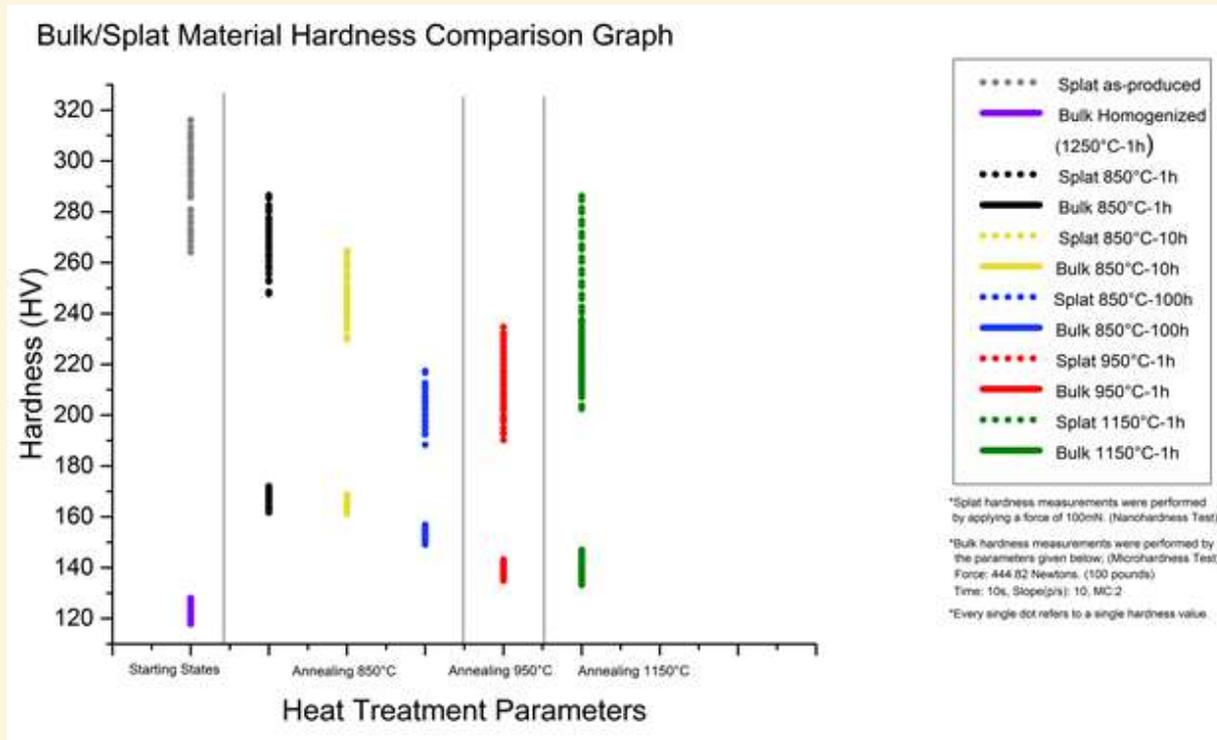
1<sup>st</sup> shell bond lengths depend on the element (Short-range order!)

- Lowest for Co (2.510(2) Å, extrapolation for  $T < 50\text{K}$ )
- Highest for Cu (2.536(3) Å, extrapolation for  $T < 50\text{K}$ )

$\sigma^2$  for Ni-1<sup>st</sup> shell is half the value compared to that of Cr, Fe and Co

- 1<sup>st</sup> shell around Ni more ordered (highest at.% of Ni in CCA a possible reason)
- Noisy spectra at Cu-K edge do not allow reliable  $\sigma^2$  refinement

# Results: hardness



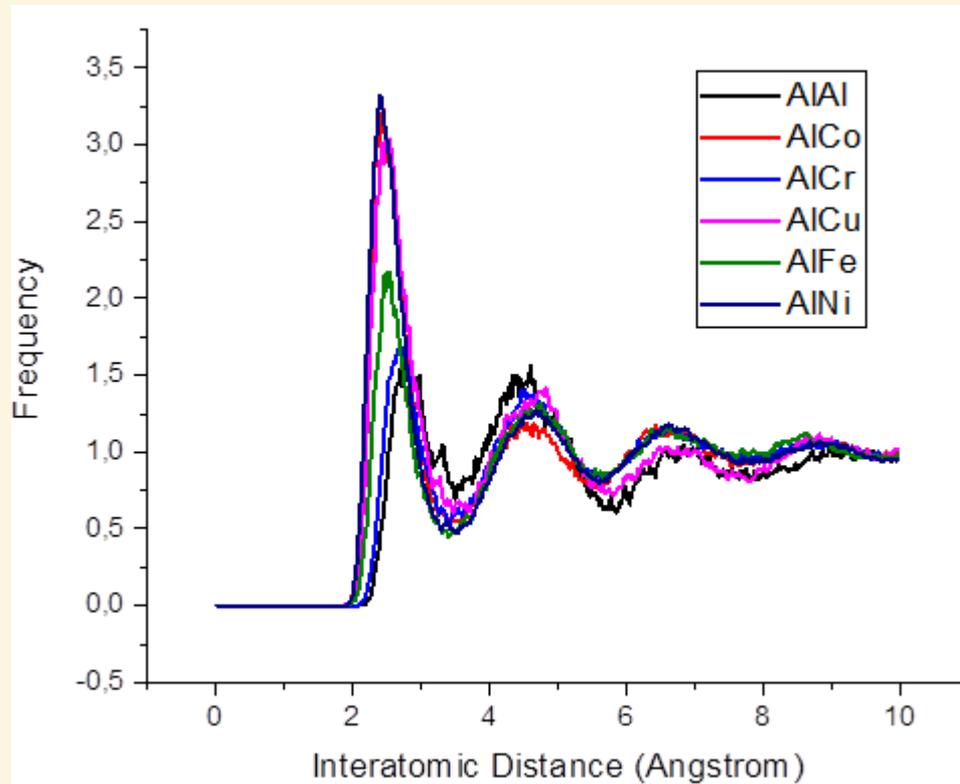
- Hardness of splats higher than hardness of bulk specimens
- Data scattering on splats higher than on bulk specimens
- Decrease in hardness at 850 °C with increasing annealing time

Two-hardening mechanisms

$T > 900^{\circ}\text{C}$  : short-range order in single phase region

$T < 900^{\circ}\text{C}$  : precipitation mechanism in  $\gamma/\gamma'$  phase region

# Results: ab initio molecular dynamics



- Crystalline state not reached at 1000°C (still liquid state)
- Liquid state reveals pronounced Short-Range Order
- Al prefers Ni/Co/Cu as nearest neighbors

	Al	Cr	Fe	Co	Ni	Cu
R(Å)	1.43	1.25	1.24	1.25	1.25	1.28

## Soft X-ray absorption spectroscopy:

Ni electronic structure seems to be different than the corresponding Ni pure metal



Further studies and measurements (e.g. at Al and Cr L-edges) needed

## Extended X-ray absorption fine structure:

- 1st shell around Co and Cu atoms are the smaller and the bigger, respectively.
- 1st shell around Ni is the most ordered



Al-K edge spectrum still missing

Any relation to Ni-electronic structure?

## Hardness

- Depends on temperature and annealing time
- Depends on specimen nature (bulk vs splat)



Two hardening mechanisms:

- SRO increase (single phase)
- Precipitation ( $\gamma/\gamma'$  region)

Splat vs bulk CCAs in XAS?

## Ab initio molecular dynamics

Preferred nearest neighbors for Al (Ni, Co, Cu)



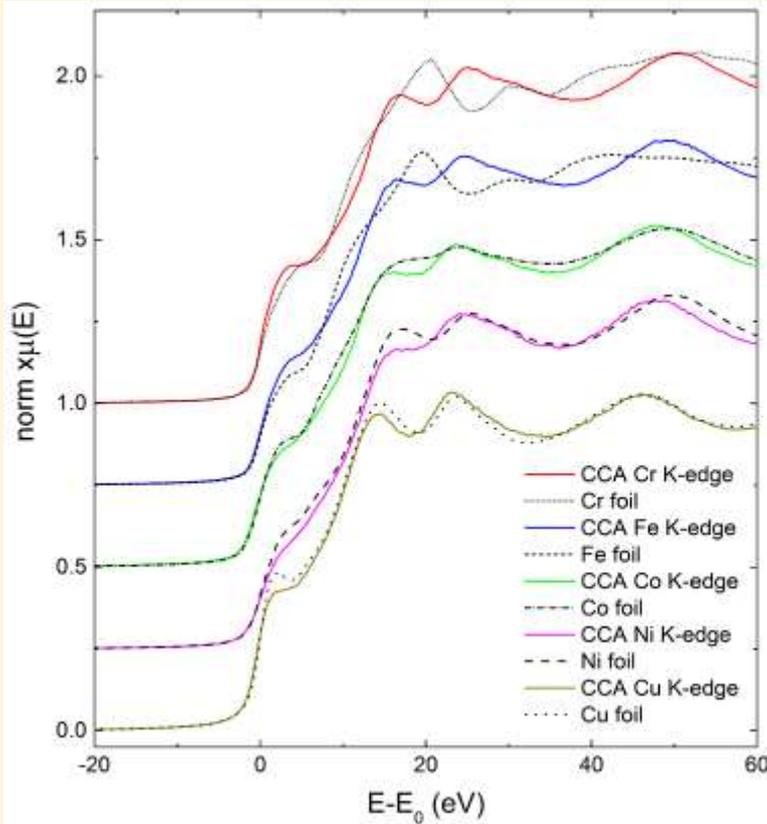
To be proved experimentally (e.g. XAS on Al-K edge)

# Acknowledgements

- Prof. Dr. John Banhart, as principal investigator of the project
- Prof. Dr. Gerhard Schumacher and Dr. A. Manzoni as additional and helpful project investigators
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- A. Öner, C. T. Cakir, for the hardness measurements
- Dr. Y. Liu, for molecular dynamics simulations

Thank you for your attention!

# Results: K-edge X-ray absorption near edge structure (XANES)



*XANES spectra for Cr, Fe, Co, Ni and Cu K-edges at 20 K in the CCA. Black lines represent reference foils.*

Pure metals and CCA:

K-edge split into two transitions

$1s \rightarrow$  mixed state (4s-4p-3d)

$1s \rightarrow$  4p state

- Fe, Cr metals have bcc structure, Co metal hcp structure. Metals spectra not easily comparable with CCA spectra



Study of e.g. 2-component (FeNi) and 3-component (CoFeNi) alloys with fcc structure is necessary