



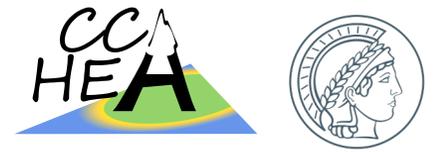
Stability and interstitial doping in high-entropy alloys from ab initio – Project: “MULTI-TRIP CCAs”

Ab initio calculations: Yuji Ikeda\*, Fritz Körmann

Experimental counterpart: Yan Ma, Chang Liu, and Dierk Raabe

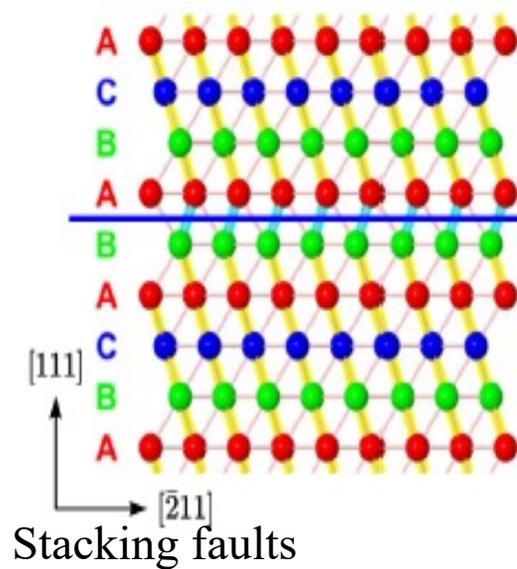
April 2021 – SPP kick-off meeting  
(\* ) now at University of Stuttgart

# Overview



- 1. Main focus in first SPP round**
- 2. Some applications FCC alloys**
- 3. Some applications BCC alloys**

- Main part: *Ab initio* simulations  $\rightarrow$  Stacking fault energies



Varying composition



Impact of interstitials (C, N, ...)



Impact of finite-temperature contributions

# FCC Alloys (mainly Cantor-family)



## Previous investigations

- **Stacking fault energies** → screening for TRIP-HEAs [1]
- **Impact interstitials on SFE (C, N) SFE [2]**
- Lattice distortions [3] and short-range order (VCoNi)

## Next steps

- **MULTI-TRIP CCAs** : dual phase HEAs with fcc->hcp + hcp->fcc transformation  
(computing generalized stacking fault energy surface, ...)
- Continue SFEs and interstitial calculations
- (short-range order and coupling to SFEs)

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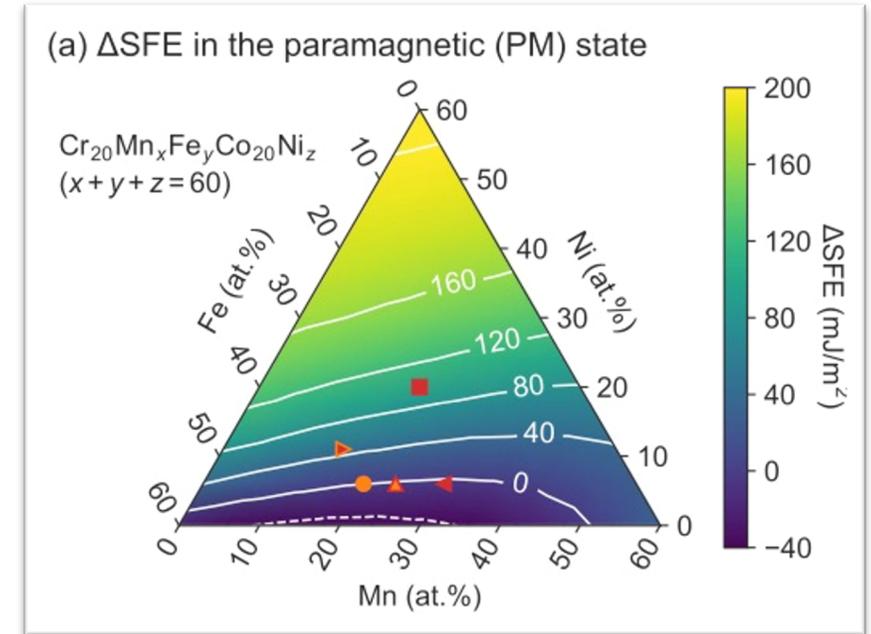


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[1] Wu, Li, Rao, Ikeda, Körmann, Dutta, Neugebauer, Raabe, Phys. Rev. Mat. 4, 033601 (2020)

**Ab initio SFE maps → new TRIP-HEAs**

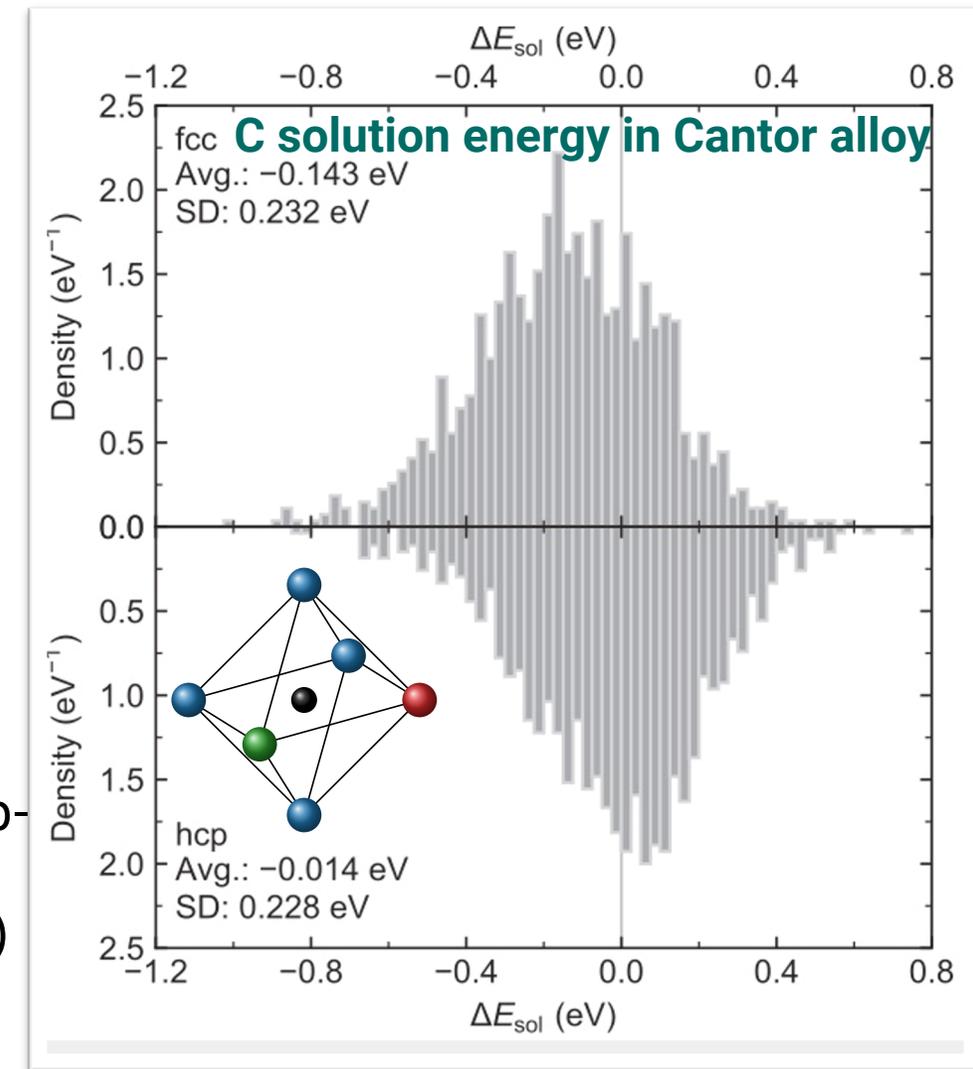
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[2] Ikeda, Tanaka, Neugebauer, Körmann, Phys. Rev. Mat. 3, 113603 (2019)  
**C increase SFE for Cantor, but decreases for Al<sub>0.5</sub>CoFeMnNi** [Kies, Hallstedt, Körmann, Haase, et al., Scripta Mat. 178, 366 (2020)].

# FCC Alloys (mainly Cantor-family)

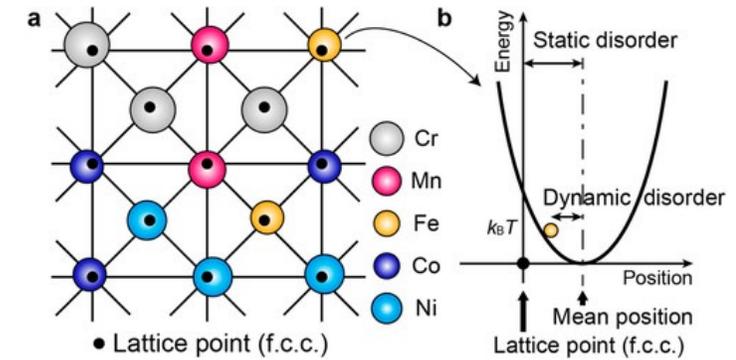
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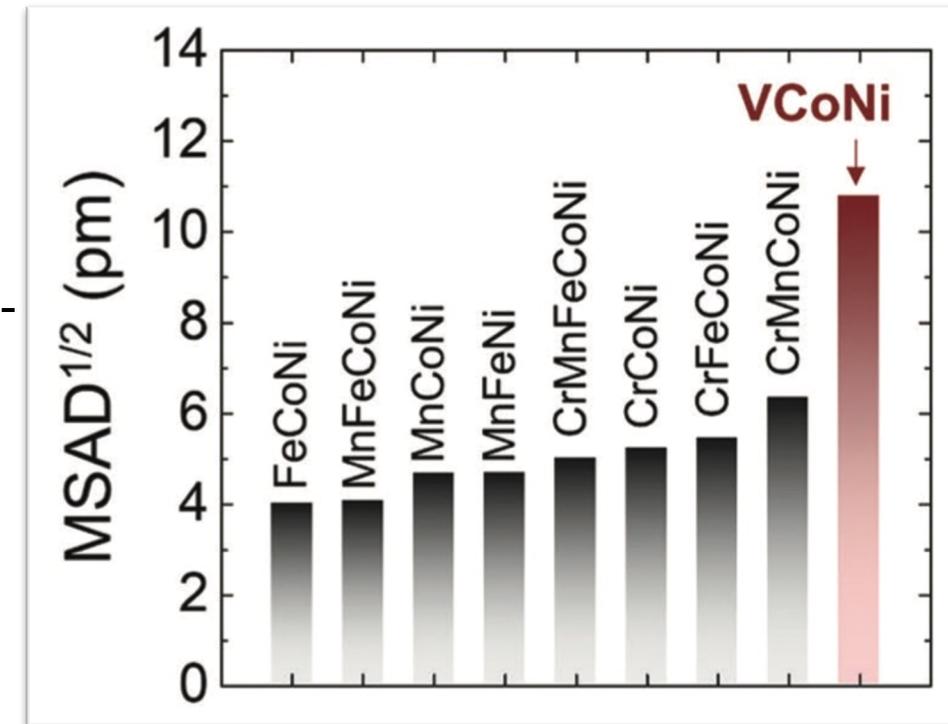
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- Continue SFEs and interstitial calculations
- (short-range order and coupling to SFEs)

## Lattice distortions from DFT



Okamoto et al. (2016)



[3] Sohn, Körmann, Raabe, *et al.*,  
Adv. Mat. 31, 1807142 (2019)

# FCC Alloys (mainly Cantor-family)



## Previous investigations

- Stacking fault energies → screening for
- Impact interstitials on SFE (C, N) SFE [2]
- Lattice distortions and short-range order

## Next steps

- **MULTI-TRIP CCAs**: dual phase HEAs with  $\gamma$  →  $\epsilon$  transformation (computing generalized stacking fault energies)
- Continue SFEs and interstitial calculations
- (short-range order and coupling to SFEs)

### COMMUNICATION

Alloys

ADVANCED MATERIALS  
www.advmat.de

## Bidirectional Transformation Enables Hierarchical Nanolaminate Dual-Phase High-Entropy Alloys

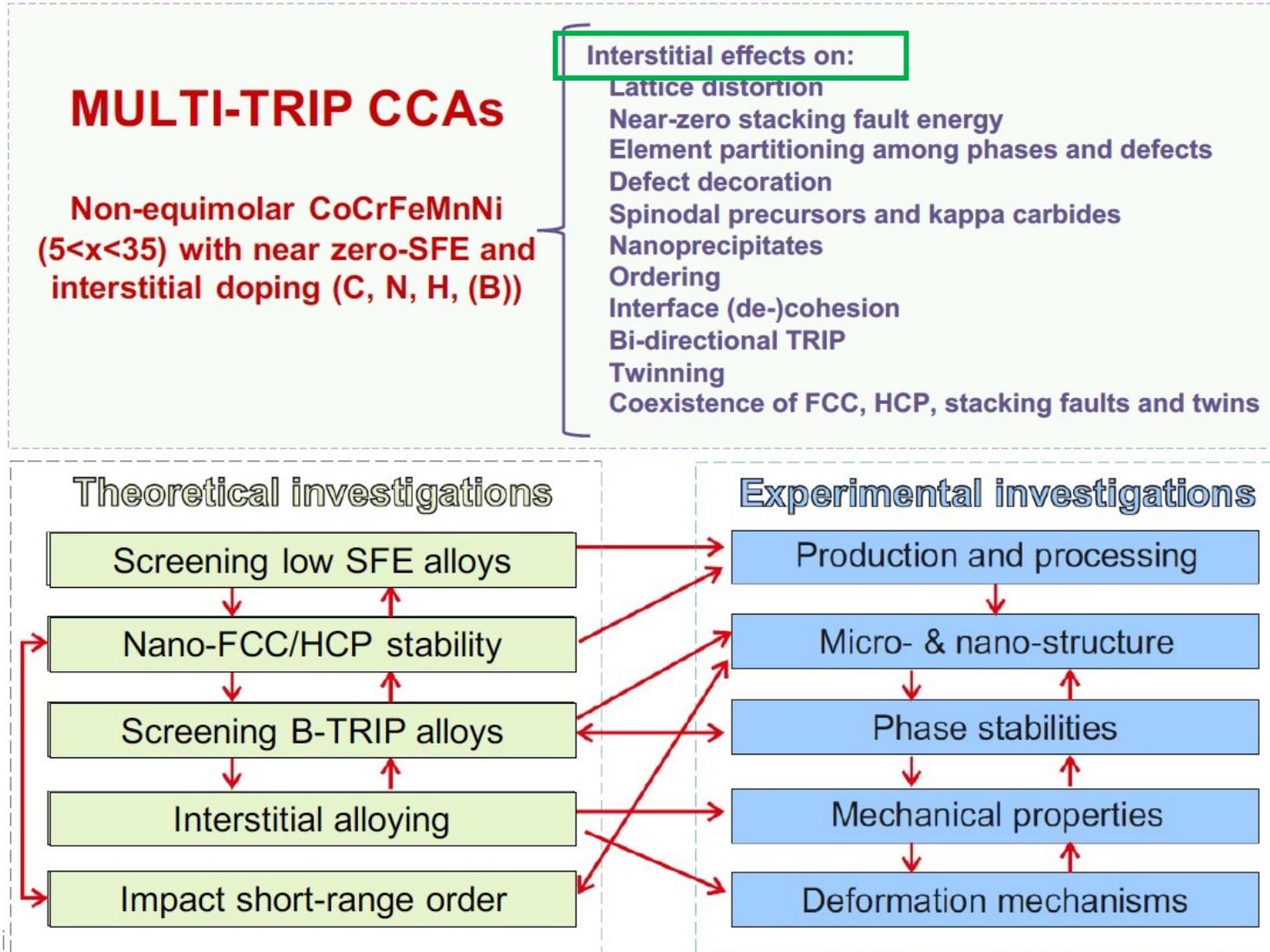
Wenjun Lu, Christian H. Liebscher, Gerhard Dehm, Dierk Raabe, and Zhiming Li\*

### FCC Fe<sub>50</sub>Mn<sub>30</sub>Co<sub>10</sub>Cr<sub>10</sub>

“This is achieved by triggering both, dynamic forward transformation from a face-centered-cubic matrix into a hexagonal-close-packed nanolaminate structure and the dynamic reverse transformation from  $\epsilon$  into  $\gamma$ . This new mechanism is referred to as the “**bidirectional transformation induced plasticity**” (B-TRIP) effect, which is enabled through a near-zero yet positive stacking fault energy of  $\gamma$ .”

→ **Can we exploit and screen for B-TRIP HEAs?**

# MULTI-TRIP CCAs – Project Overview



# FCC Alloys (mainly Cantor-family)



## Previous investigations

- **Stacking fault energies** → screening for TRIP-HEAs [1]
- **Impact interstitials on SFE (C, N) SFE** [2]
- Lattice distortions and short-range order (VCoNi) [3]

## Next steps

- **MULTI-TRIP CCAs**: dual phase HEAs with fcc->hcp + hcp->fcc transformation  
(computing generalized stacking fault energy surface, ...)
- Continue SFEs and interstitial calculations
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# BCC Alloys



## Previous investigations

- Phase stability + short-range order, e.g., BCC NbMoTaW [4], AlNbTiV [5],
- omega-BCC-HCP stability in Ti-HEAs

## Next steps

- Continue short-range order / stability, e.g., B2 phase in Al-containing HEAs

# BCC Alloys

DFT → on-lattice machine learning potentials →  
Monte Carlo simulations



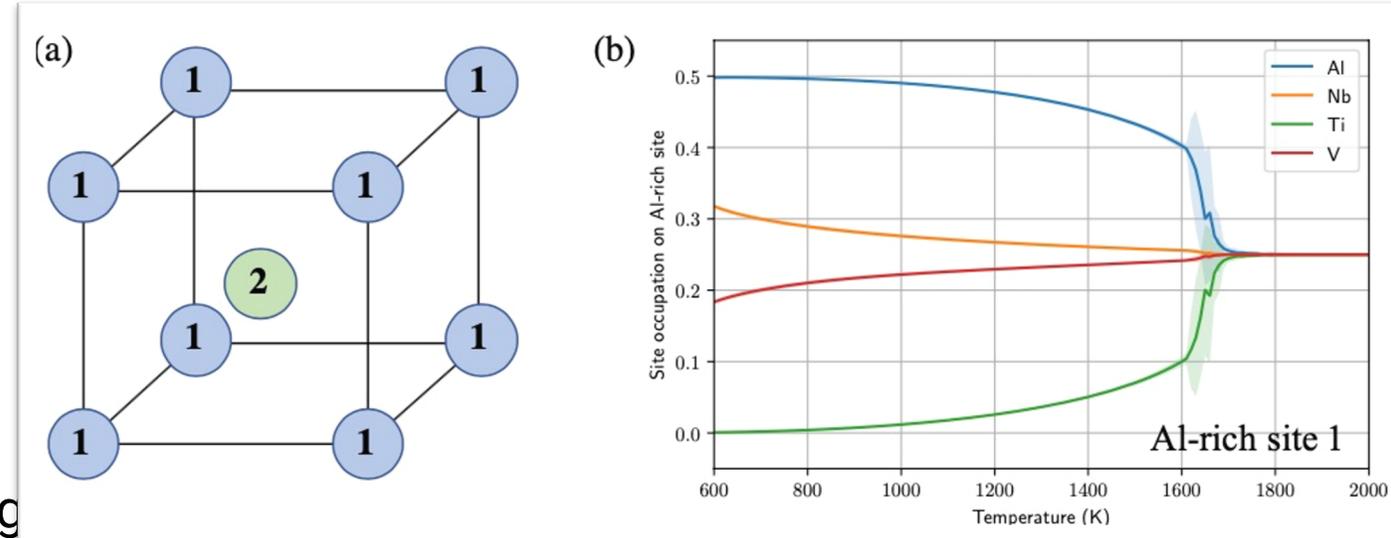
Bcc AlNbTiV → B2 ordering (and sublattice disorder)

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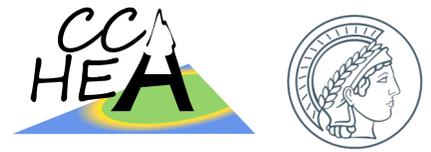
## Next steps

- Continue short-range order / stability, e.g. B2 phase in Al-containing HEAs



[5] Körmann et al., (under review)

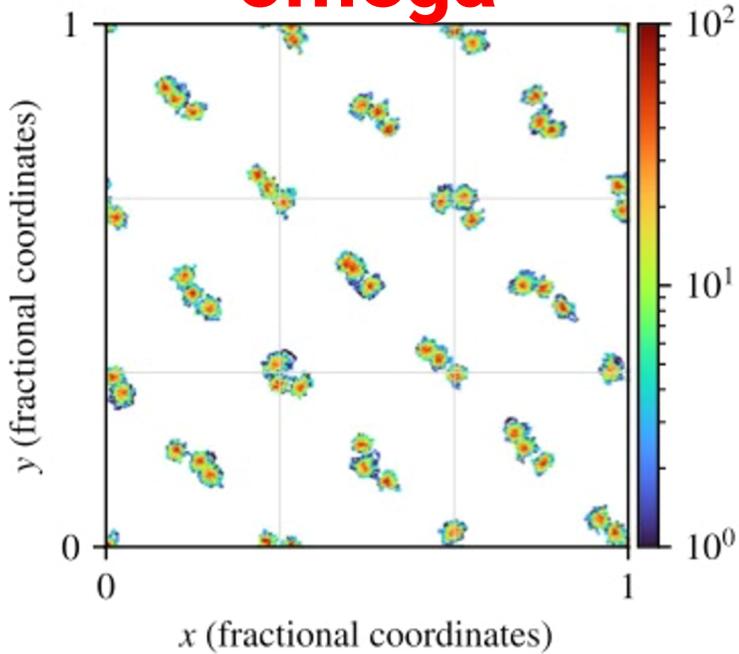
# BCC Alloys



## Example: TiZrHf

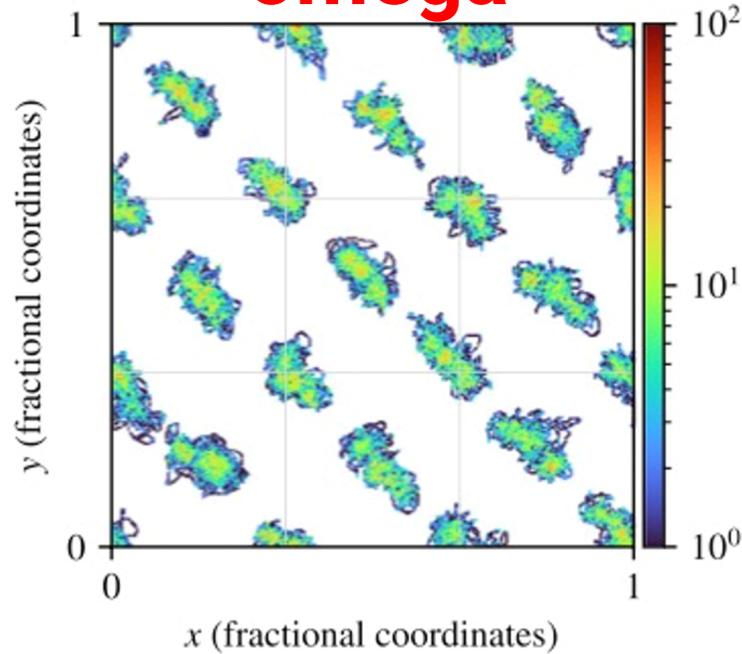
**a** Trajectory of atomic positions at 300 K

**omega**



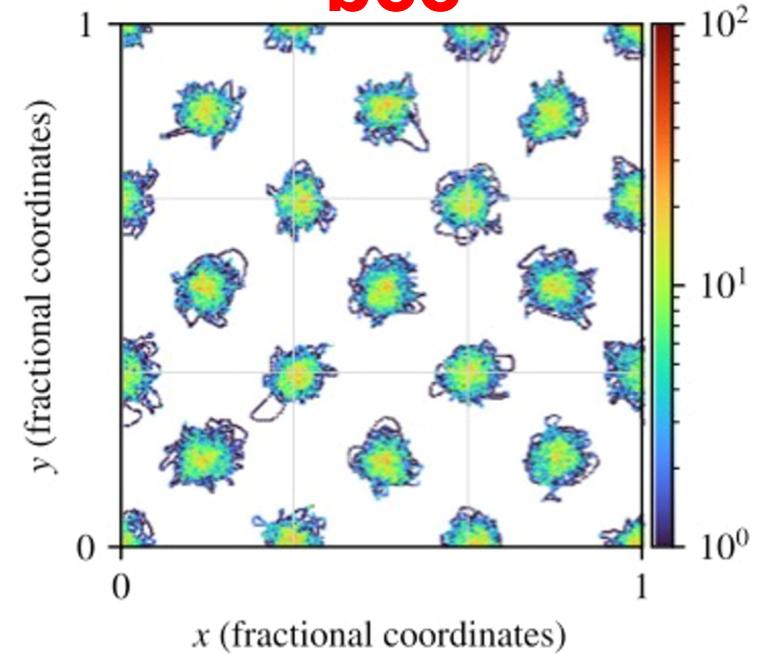
**b** Trajectory of atomic positions at 900 K

**omega**



**c** Trajectory of atomic positions at 1500 K

**bcc**



[5] Ikeda, Gubaev, Neugebauer, Grabowski, Körmann, npj Comp Mat **34**, 1 (2021)

# BCC Alloys



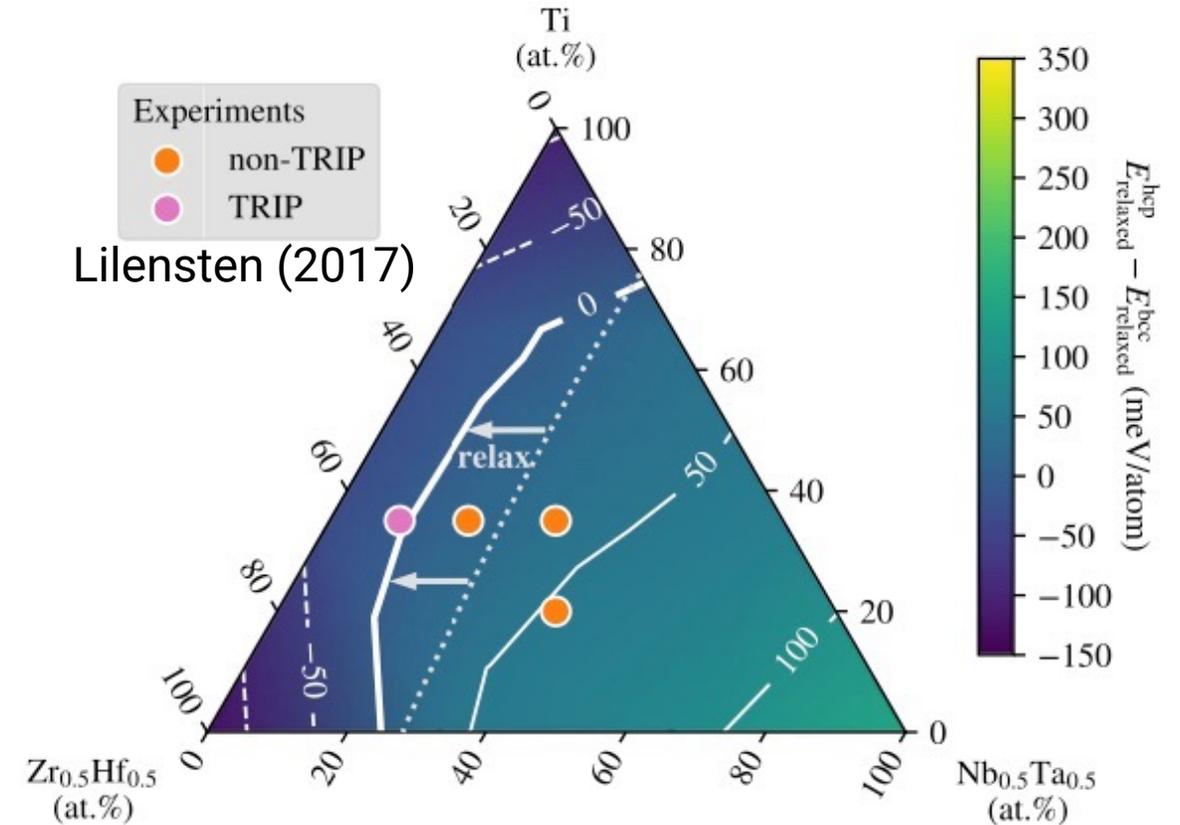
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- **omega-BCC-HCP stability in Ti-HEAs**

## Next steps

- Continue short-range order / stability, e.g., B2 phase in Al-containing HEAs

## TiZrHfNbTa: BCC – HCP stability (T=0K)



[5] Ikeda, Gubaev, Neugebauer, Grabowski, Körmann, npj Comp Mat **34**, 1 (2021)

**Thank you for your  
attention!**

**MAX-PLANCK-INSTITUT**  
FÜR EISENFORSCHUNG GmbH

