

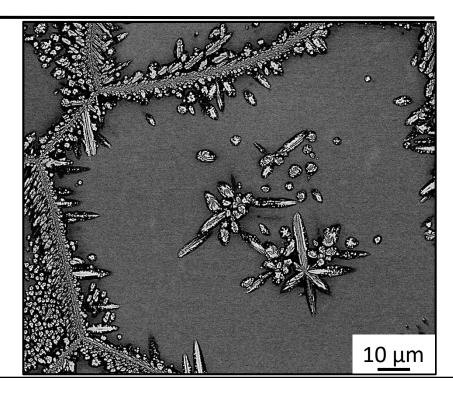


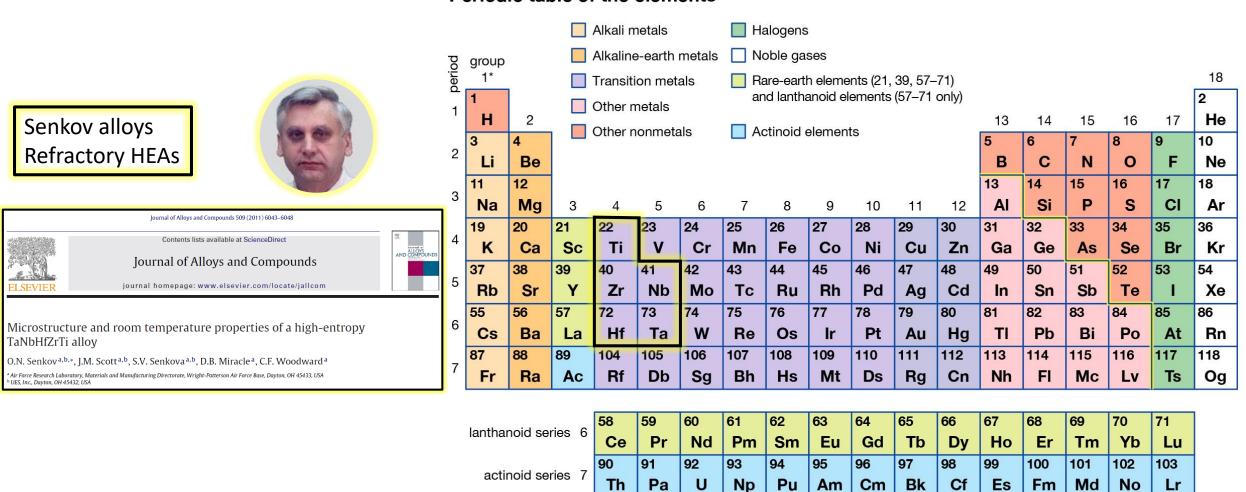
Phase stability, precipitation kinetics, nanoscale elemental distributions and their effect on tensile properties in refractory TiZrNbHfTa BCC high-entropy alloys Part I

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Periodic table of the elements

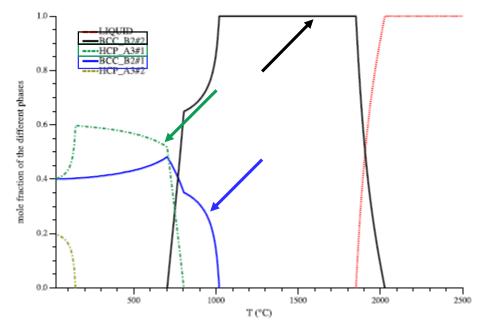
*Numbering system adopted by the International Union of Pure and Applied Chemistry (IUPAC). © Encyclopædia Britannica, Inc.

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Phase stability of TiZrNbHfTa



CALPHAD calculations of the TiZrNbHfTa alloy with the TCHEA1 database, evolution of the mole fraction of the stable phases

Acta Materialia 142 (2018) 201-212



Full length article

Thermodynamic instability of a nanocrystalline, single-phase TiZrNbHfTa alloy and its impact on the mechanical properties

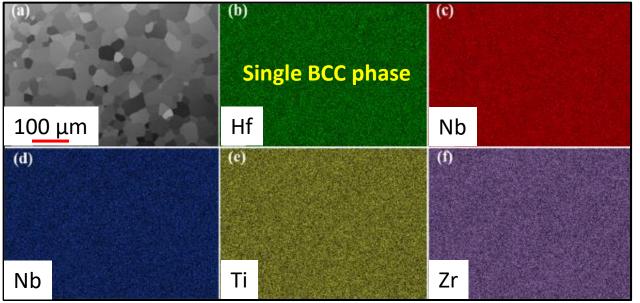


B. Schuh ^{a, *}, B. Völker ^a, J. Todt ^b, N. Schell ^c, L. Perrière ^d, J. Li ^e, J.P. Couzinié ^d, A. Hohenwarter ^a

HEA More than the state of the

Cold rolled (70% reduction) \longrightarrow homogenized at 1200°C for 5 min and water quenched

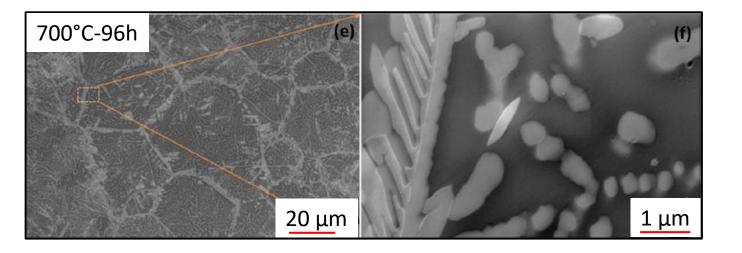
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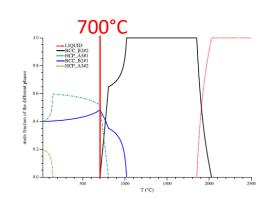


	Scripta Materialia 158 (2019) 50-56
	Contents lists available at ScienceDirect
5-8 (4)	Scripta Materialia
ELSEVIER	journal homepage: www.elsevier.com/locate/scriptamat

Phase transformations of HfNbTaTiZr high-entropy alloy at intermediate temperatures☆



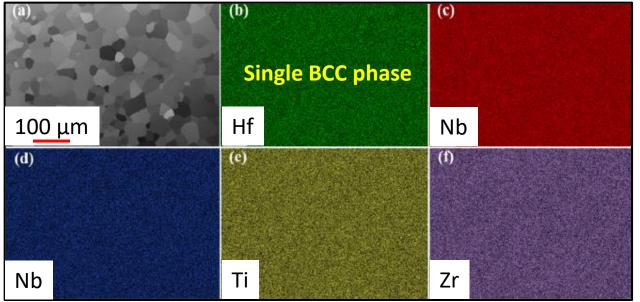




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Cold rolled (70% reduction) \longrightarrow homogenized at 1200°C for 5 min and water quenched

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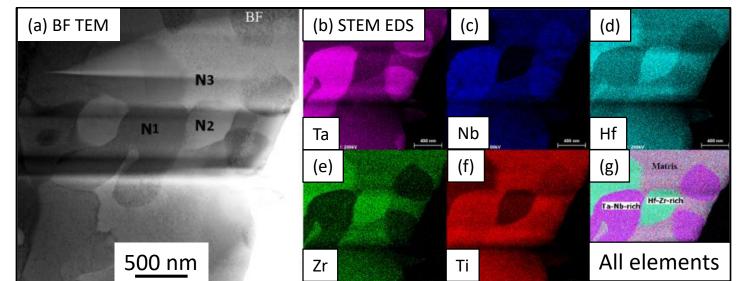




Phase transformations of HfNbTaTiZr high-entropy alloy at intermediate temperatures☆



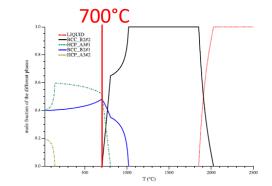
S.Y. Chen^a, Y. Tong^a, K.-K. Tseng^b, J.-W. Yeh^b, J.D. Poplawsky^c, J.G. Wen^d, M.C. Gao^{e,f}, G. Kim^g, W. Chen^g, Y. Ren^h, R. Feng^a, W.D. Li^a, P.K. Liaw^{a,*}



N1: Ta-Nb rich phase, BCC

N2: Hf-Zr rich phase, HCP

N3: matrix, BCC

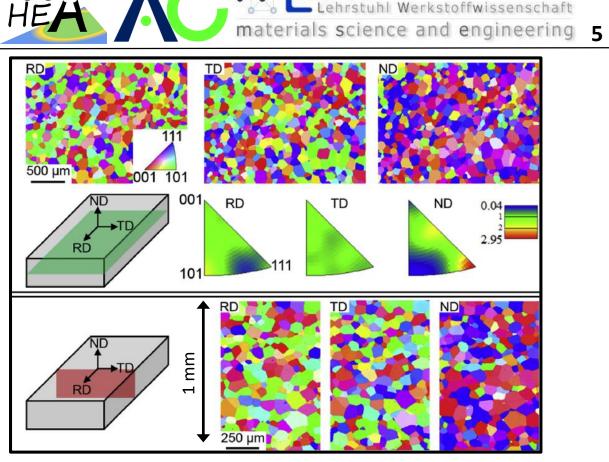


as-received state

cold rolling (~80%) + annealing at 1100°C for 5h

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Laplanche et al., Journal of alloys and compounds 799 (2019) 538-545



J.-P. Couzinié Institut de Chimie et des Materiaux Paris-Est

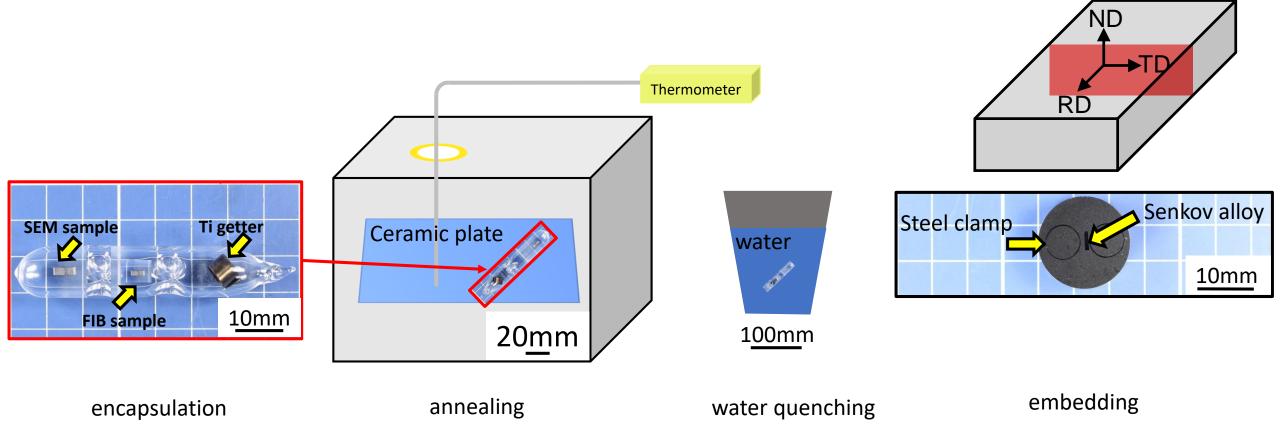
- Nearly equiaxed grains
- Nearly no texture along RD and TD
- Slight <111>- and <001>-fiber textures along ND
- average grain size ~80 μm

Experiments

- Sheets (thickness ~1mm) recrystallized at 1100°C for 5 h
- Specimens (4mm × 2mm × 1mm) encapsulated in evacuated-quartz tubes with a Ti getter (thin coiled sheet)

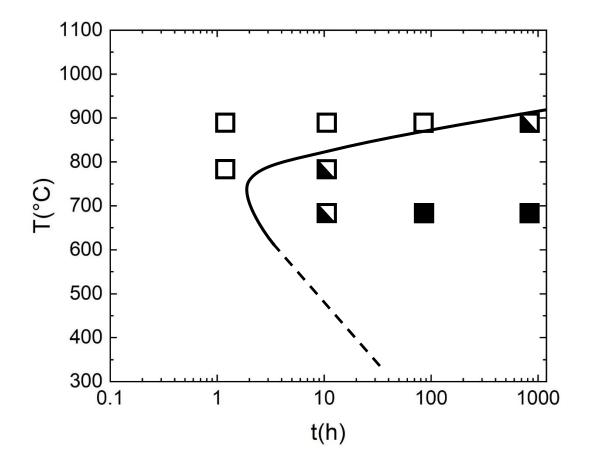
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- Long-term anneal + water quenching.
- The annealed specimens were then embedded in epoxy with a steel clamp to observe the microstructure along a direction parallel to RD.



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Preliminary TTT diagram



Temperature						
700°C	1h	10h	100h	1000h		
800°C	1h	10h	100h	1000h		
900°C		10h	100h	1000h		

° '

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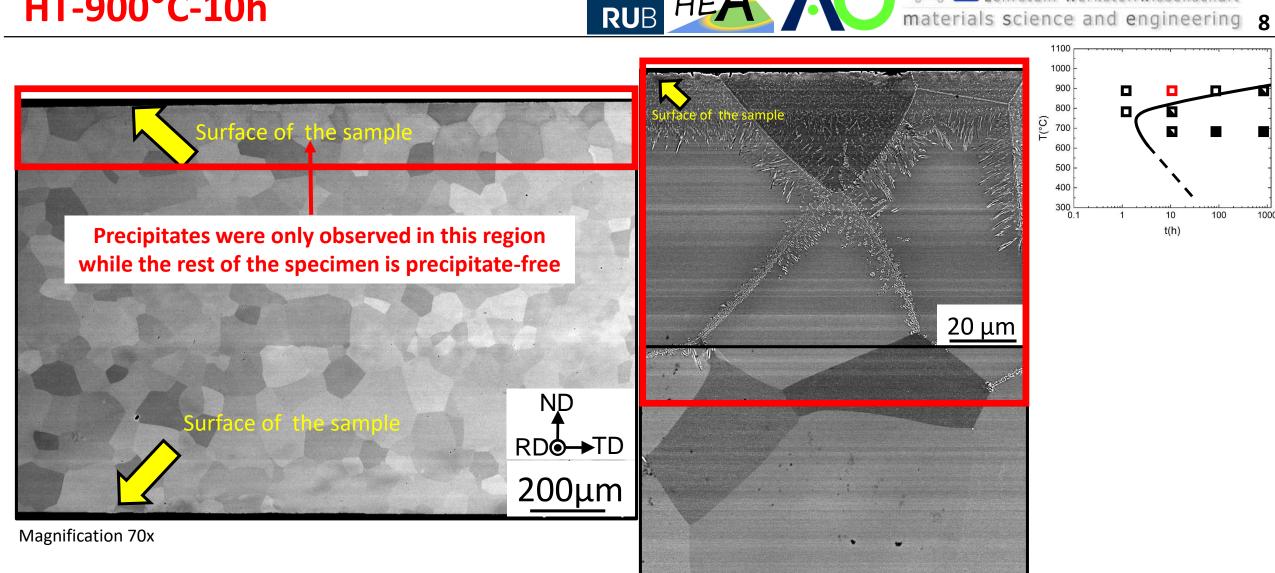
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done in the plan

- without precipitates
- with precipitates
- precipitates just at the GB



00

20 µm

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100

1000

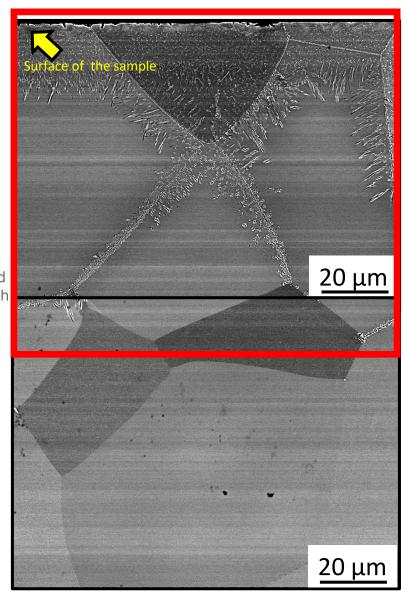
O₂ and N₂ contamination at the surface Do they induce a change in phase stability?

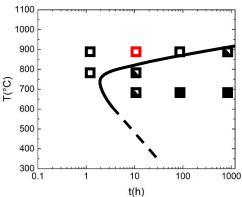
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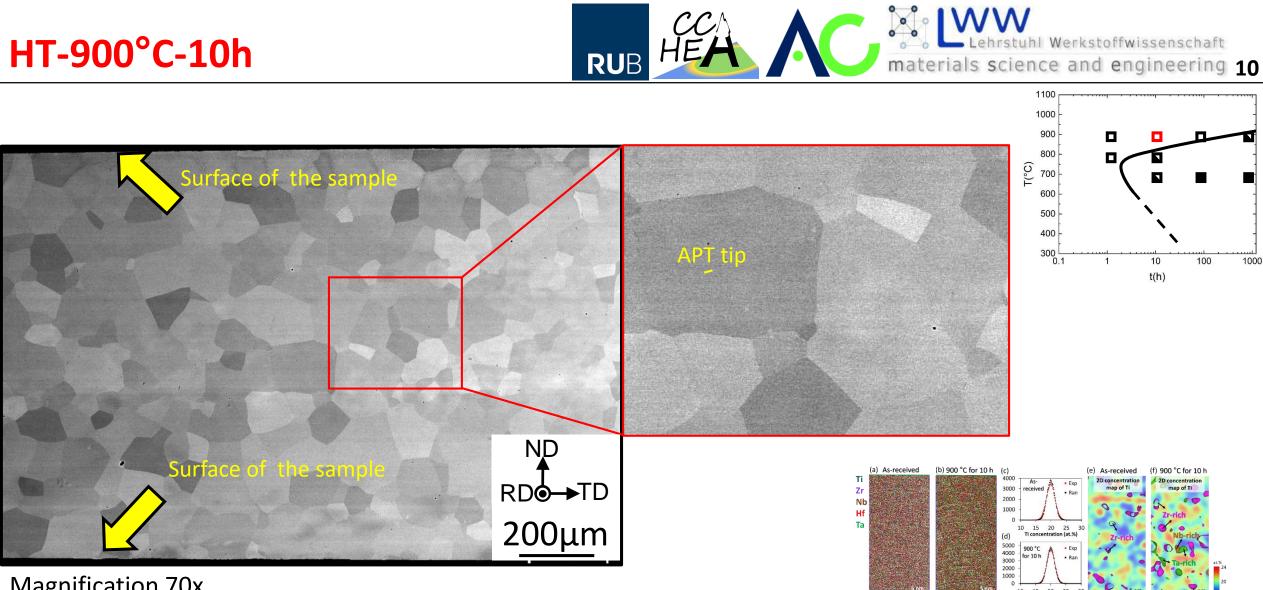
Collaboration with KIT

- Effect of O_2 and N_2 on phase stability
- Two cast and homogenized ingots will be produced
- TiZrNbHfTa
- TiZrNbHfTa +2 at.%O₂
- TiZrNbHfTa +2 at.%N₂

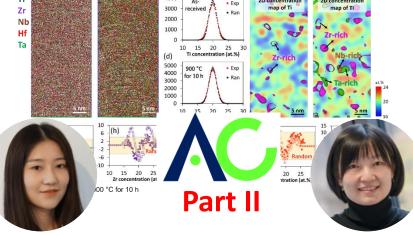






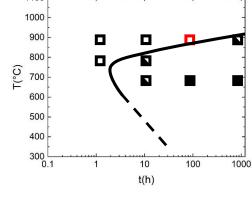


Magnification 70x

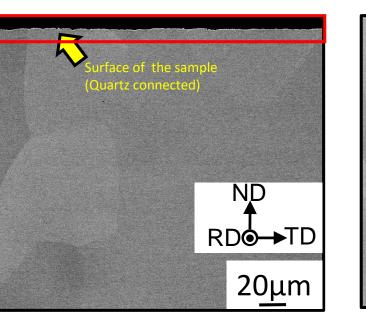


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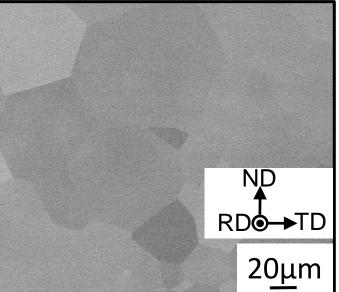








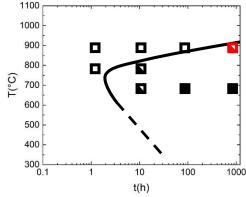


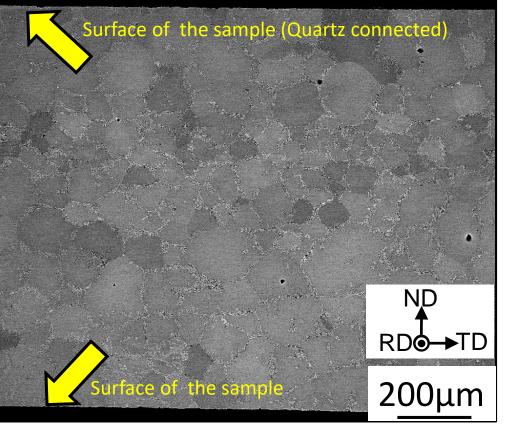


Surface of the sample (Quartz connected) no precipitate was observed ND RD**Ó→**TD ce of the sample 200µm

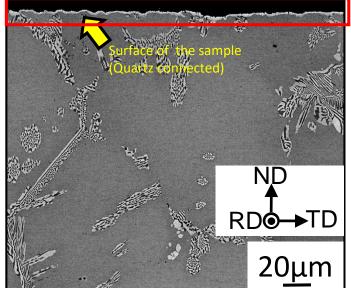
> Better vacuum in quartz tube? The layer was dissolved?



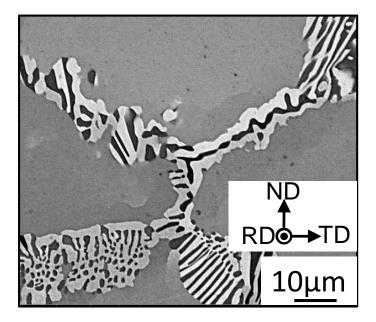




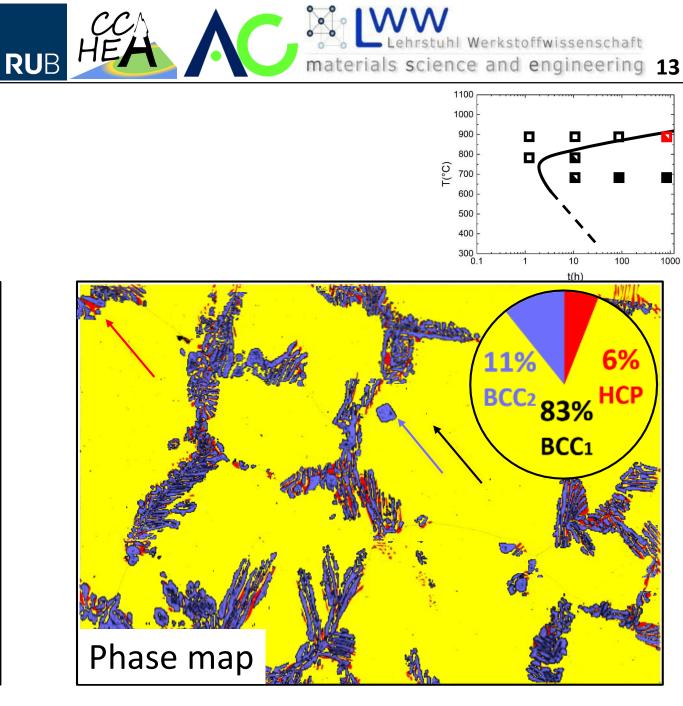
Upper part of the sheet



Middle part of the sheet

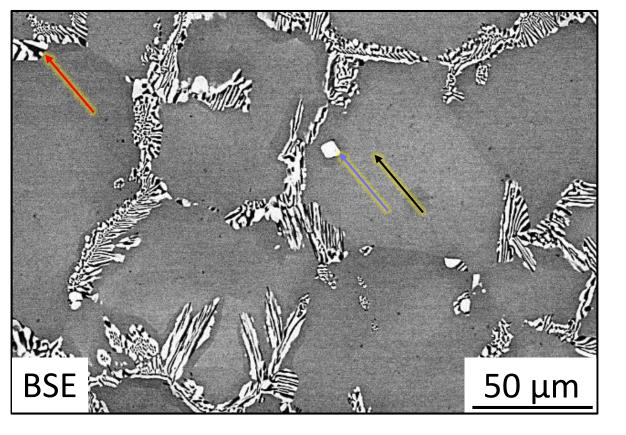


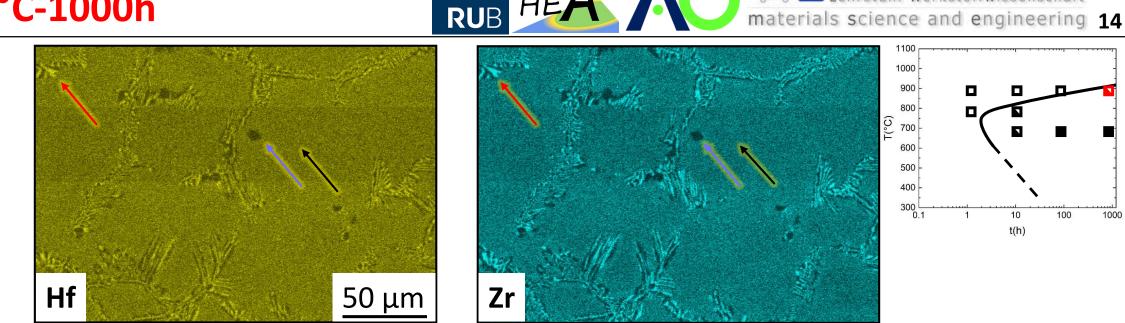
Precipitates were observed mostly at grain boundaries



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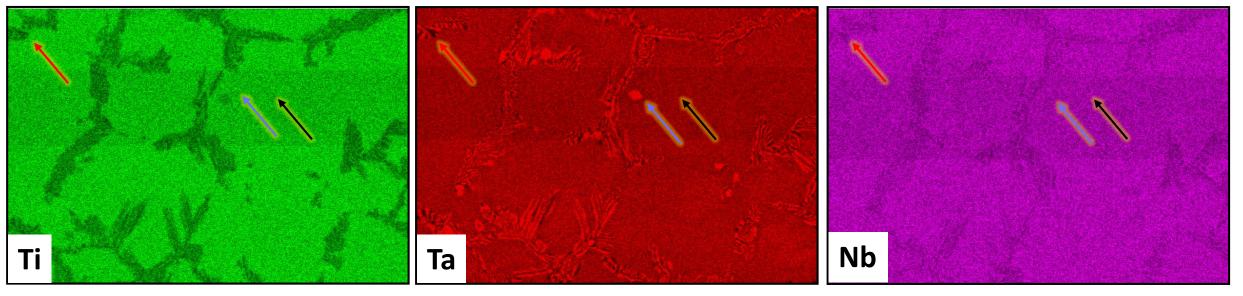


HCP-precipitates: enriched in Hf, Zr

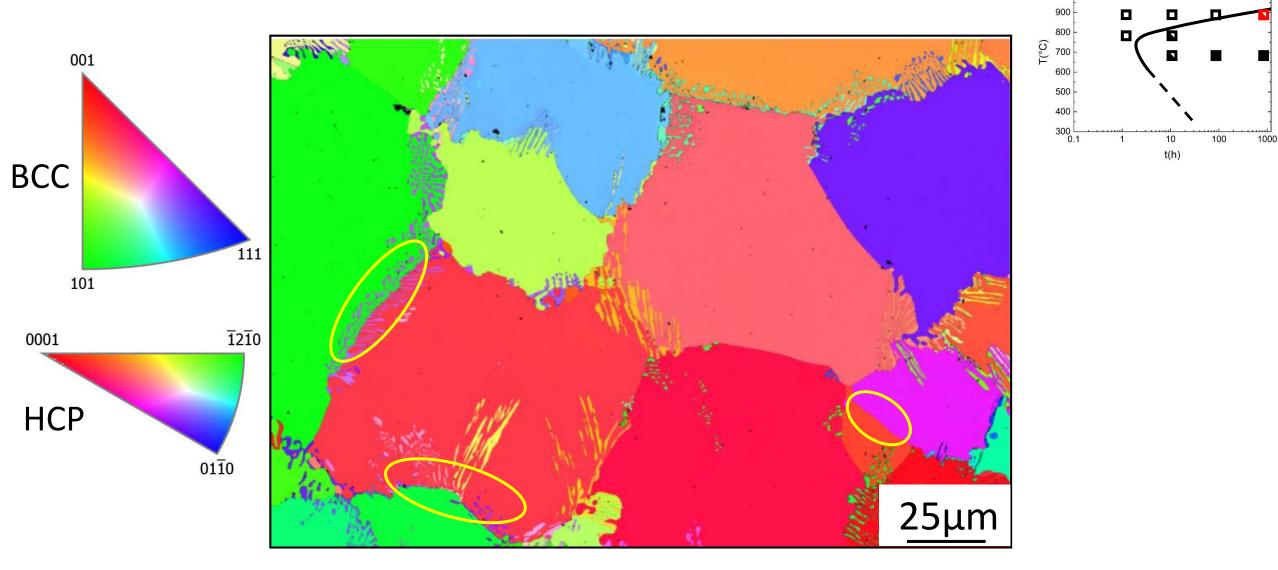
BCC precipitates: enriched in Ta

BCC-matrix: mixture, slight enrichment in Ti

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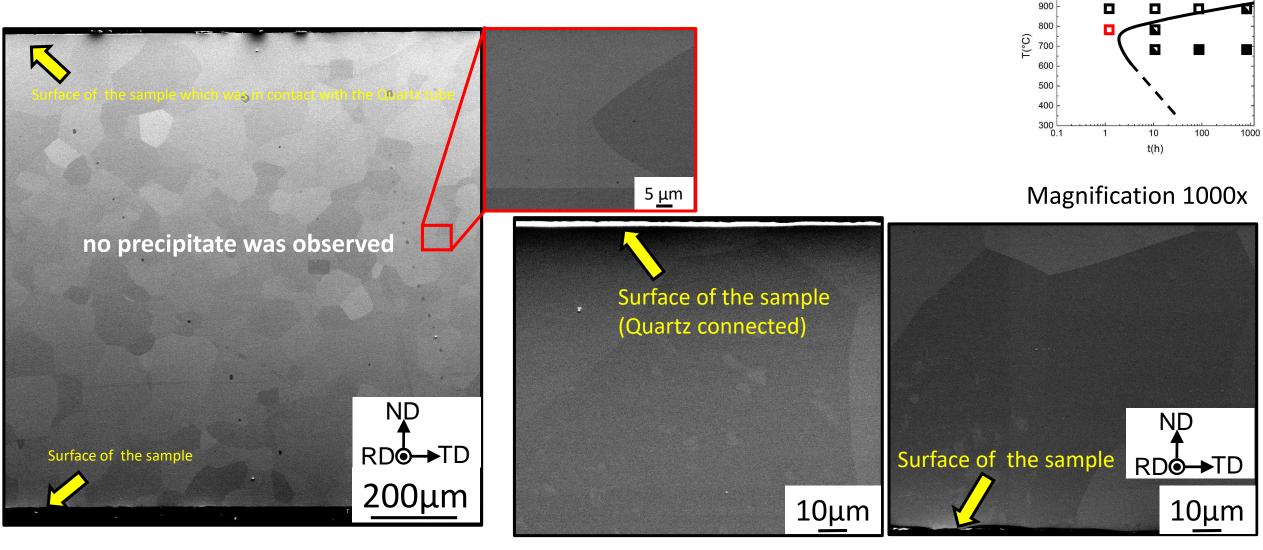




HT-800°C-1h



Magnification 70x



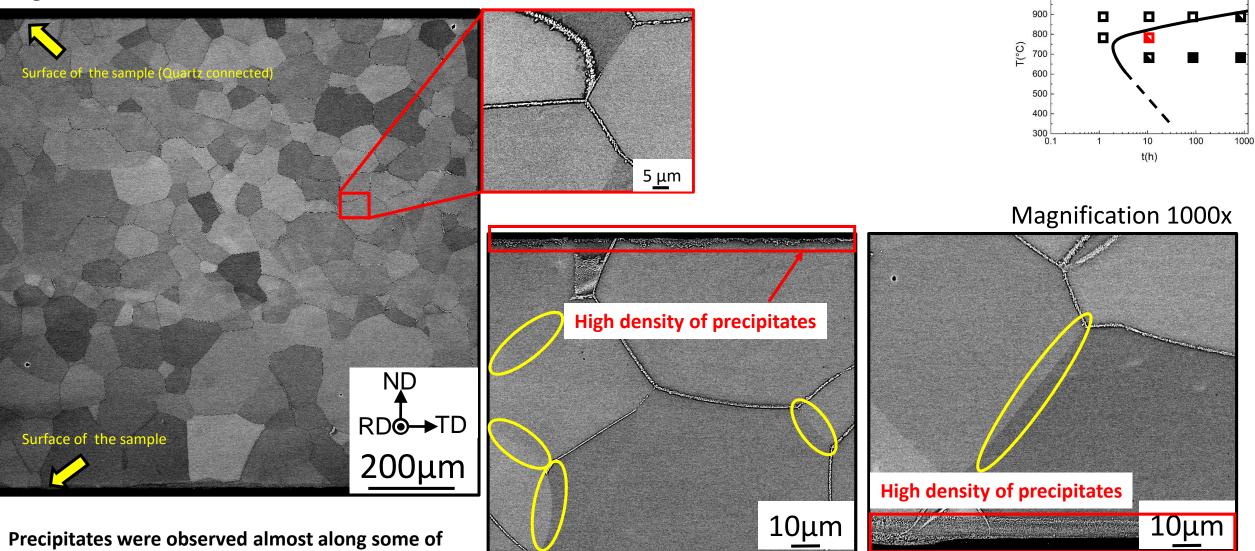
Upper part of the sheet

Bottom part of the sheet

HT-800°C-10h

the grain boundaries

Magnification 70x



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Upper part of the sheet

Bottom part of the sheet

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HT-700°C-10h

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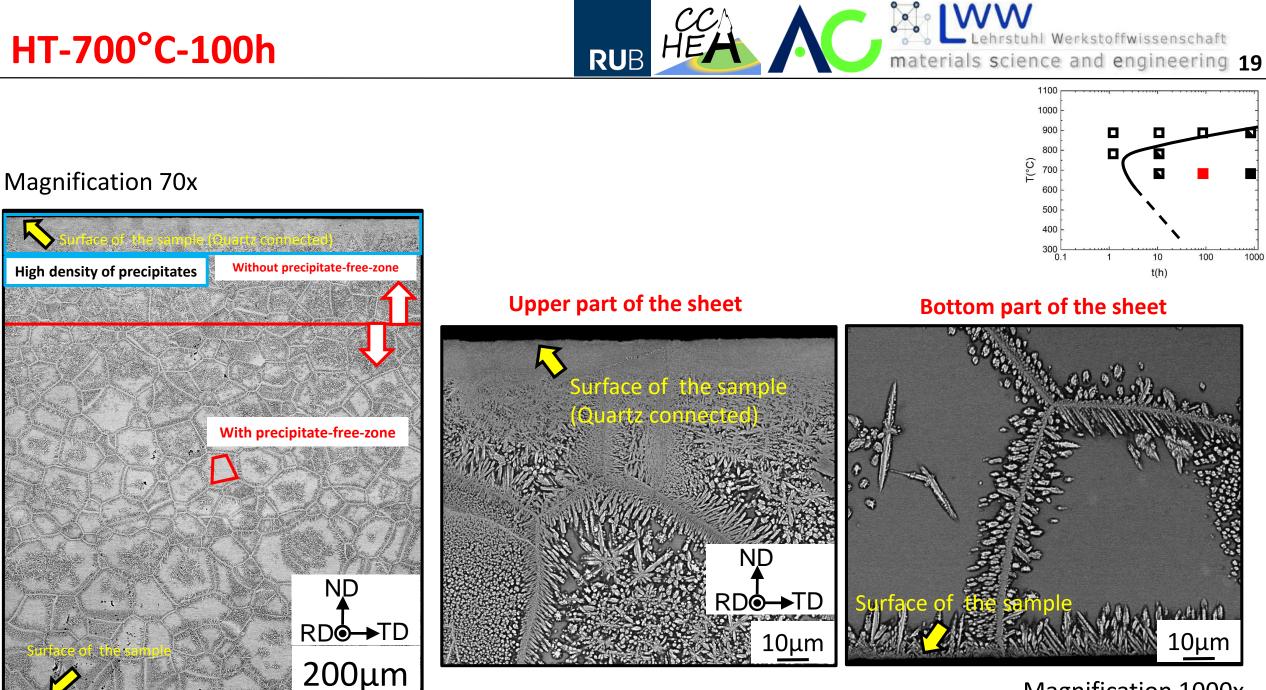
Magnification 70x 1000 900 800 T(°C) 700 Surface of the sample which was in contact with the Quartz tub 600 500 400 300 ⊾ 0.1 10 100 1000 t(h) Upper part of the sheet 5 µm Magnification 1000x **High density of precipitates** no precipitate ND RD**Ó→**TD High density of precipitates ND Surface of the sample 200µm Surface of the sam RD**Ó→**TD Second - second 10µm 10µm

Precipitates were observed almost along all the grain boundaries

Upper part of the sheet

Bottom part of the sheet

HT-700°C-100h



Magnification 1000x

100

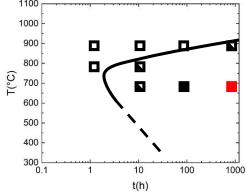
0.41.1 0

10µm

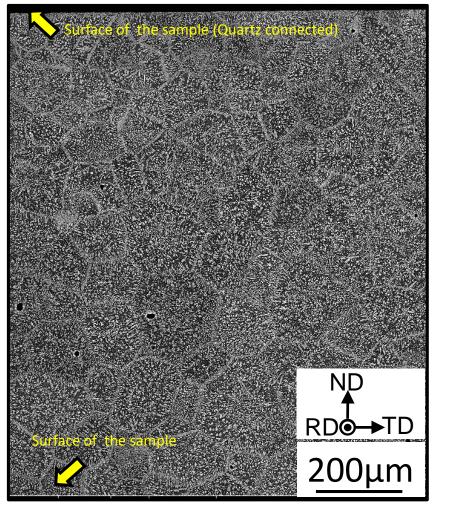
1000

NWN §

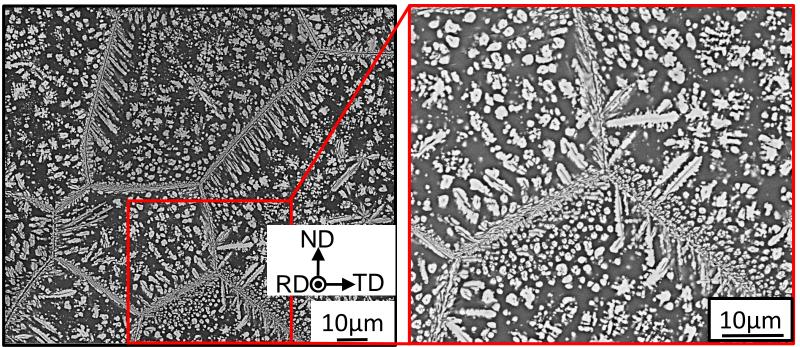




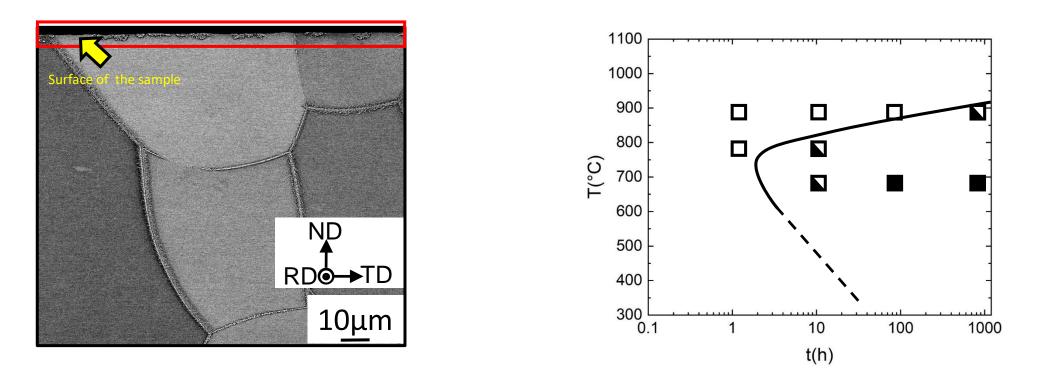
Sheet overview



Middle part of the sample



Conclusion

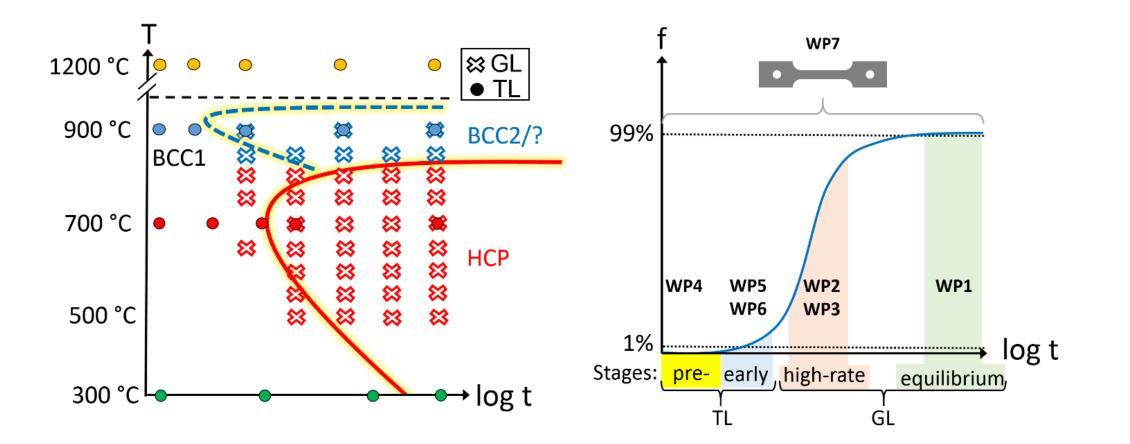


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- Even low amount of interstitial elements (contamination) like Oxygen or Nitrogen during annealing cause forming of precipitates on the surface of the annealed samples.
- At 900°C after 10h and also 100h no precipitate was observed, but after 1000h precipitates were observed.
- At 800°C after 1h, no precipitate was observed but after 10h the precipitates could be seen once again along the grain boundaries.
- At 700°C after 10h, precipitates were observed along almost all the grain boundaries and after 100h there were present also within the grains.

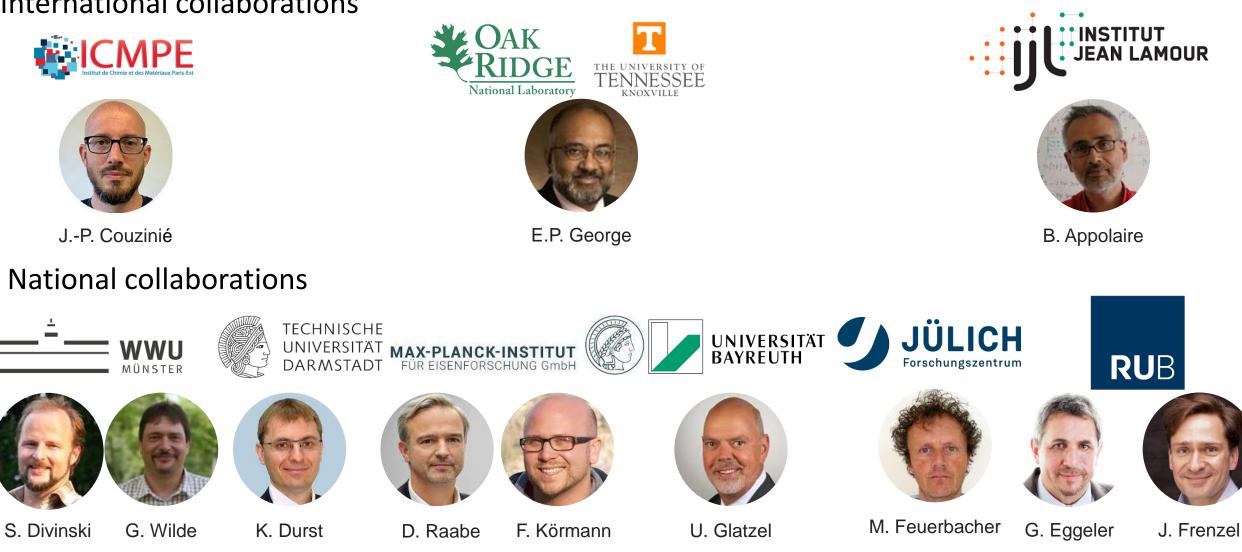
Mechanical tests in the future



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Collaborations

International collaborations

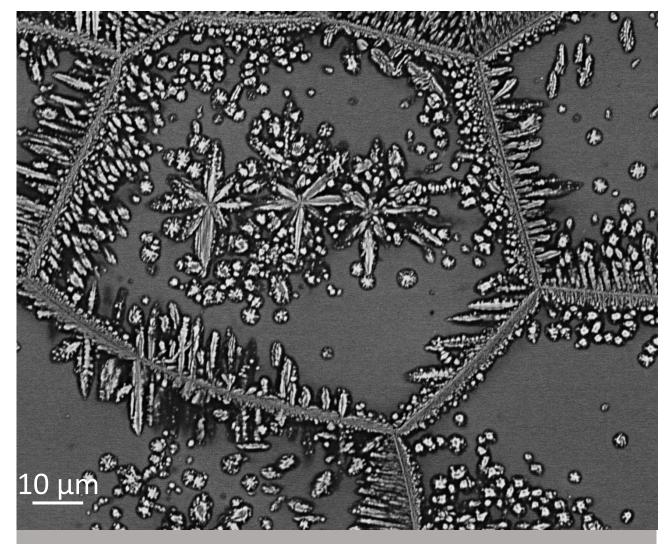


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Thank you for your attention